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(54) 【発明の名称】 硬質被覆層がすぐれた耐チッピング性を有する表面被覆超硬合金製切削工具

(57)【要約】

【課題】 硬質被覆層がすぐれた耐チッピング性を有する表面被覆超硬合金製切削工具を提供する。

【解決手段】 WC基超硬合金基体の表面に、A12O。 層を含む硬質被覆層、例えばTiC層、TiN層、TiCN層、TiO2層、TiCO層、TiNO層、およびTiCNの層からなるTi化合物層のうちの1種または2種以上と、A12O3層で構成した硬質被覆層を8~25μmの平均層厚で化学蒸着および/または物理蒸着してなる表面被覆超硬合金製切削工具において、前記硬質被覆層を構成するA12O3層を、上側部と下側部接相互の長さ方向の断面形状および断面寸法が均等化した縦長成長の柱状単一化多面体結晶の並列集合組織、上記下側部を、隣接相互の長さ方向の断面形状および断面寸法が均等化した縦長成長の柱状単一化多面体結晶の並列集合組織、上記下側部を、隣接相互の長さ方向の断面形状および断面寸法に大きな相異がある縦長成長の柱状多様化多面体結晶の並列集合組織、を有するA12O3層で構成する。

【特許請求の範囲】

【請求項1】 炭化タングステン基超硬合金基体の表面 に、酸化アルミニウム層を含む硬質被覆層を8~25 μ mの平均層厚で化学蒸着および/または物理蒸着してなる表面被覆超硬合金製切削工具において、前記硬質被覆層を構成する酸化アルミニウム層を

上側部と下側部(基体側)で異なる結晶組織を有し、 上記上側部を、隣接相互の長さ方向の断面形状および断 面寸法が均等化した縦長成長の柱状単一化多面体結晶の 並列集合組織、

上記下側部を、隣接相互の長さ方向の断面形状および断面寸法に大きな相異がある縦長成長の柱状多様化多面体結晶の並列集合組織、とした酸化アルミニウム層で構成したことを特徴とする硬質被覆層がすぐれた耐チッピング性を有する表面被覆超硬合金製切削工具。

【請求項2】 炭化タングステン基超硬合金基体の表面に、Tiの炭化物層、窒化物層、炭窒化物層、酸化物層、炭酸化物層、窒酸化物層、および炭窒酸化物層からなるTi化合物層のうちの1種または2種以上と、酸化アルミニウム層とで構成された硬質被覆層を8~25μmの平均層厚で化学蒸着および/または物理蒸着してなる表面被覆超硬合金製切削工具において、前記硬質被覆層を構成する酸化アルミニウム層を、

上側部と下側部(基体側)で異なる結晶組織を有し、 上記上側部を、隣接相互の長さ方向の断面形状および断 面寸法が均等化した縦長成長の柱状単一化多面体結晶の 並列集合組織、

上記下側部を、隣接相互の長さ方向の断面形状および断面寸法に大きな相異がある縦長成長の柱状多様化多面体結晶の並列集合組織、とした酸化アルミニウム層で構成したことを特徴とする硬質被覆層がすぐれた耐チッピング性を有する表面被覆超硬合金製切削工具。

【請求項3】上記酸化アルミニウム層の上側部と下側部のそれぞれの表面に対する $Cu\kappa\alpha$ 線を線源として用いたX線回折で、酸化アルミニウムのピークが現れる3 1.1度および37.7度の回折角(2θ)でのピーク高さが、37.7度の回折角に現れるピーク高さに比して31.1度の回折角に現れるピーク高さの方が相対的に高く、かつそのピーク高さの高低差が大きいことを特徴とする上記請求項1または2に記載の表面被覆超硬合金製切削工具。

【発明の詳細な説明】

[0001]

【発明の属する技術分野】この発明は、硬質被覆層がすぐれた耐チッピング性を有し、したがって例えば鋼や鋳鉄の高速切削や、高速で、かつ高送りおよび高切込みなどの重切削などの苛酷な条件で用いた場合にも切刃にチッピング(微小欠け)の発生なく、長期に亘ってすぐれた切削性能を発揮する表面被覆超硬合金製切削工具(以下、被覆超硬工具と云う)に関するものである。

[0002]

【従来の技術】従来、一般に、炭化タングステン基超硬 合金基体(以下、超硬基体という)の表面に、酸化アル ミニウム (以下、A 12 O3 で示す) 層を含む硬質被覆 層、例えばTiの炭化物 (以下、TiCで示す) 層、窒 化物(以下、同じくTiNで示す)層、炭窒化物(以 下、TiCNで示す)層、酸化物(以下、TiO,で示 す)層、炭酸化物(以下、TiCOで示す)層、窒酸化 物(以下、TiNOで示す)層、および炭窒酸化物 (以 下、TiCNOで示す)層からなるTi化合物層のうち の1種または2種以上と、 Al_2O_3 層とで構成された 硬質被覆層を10~25μmの平均層厚で化学蒸着およ び/または物理蒸着してなる被覆超硬工具が知られてお り、この被覆超硬工具が鋼や鋳鉄などの連続切削や断続 切削に用いられていることも知られている。また、上記 硬質被覆層を構成するAl₂ 〇3 層が粒状結晶組織をも つことも知られており、さらに上記A12O3層の表面 に対するCuκα線を線源として用いたX線回折で、図 6のX線回折パターンに例示されるように、A1,O。 のピークが現れる31.1度(図中A表示)および3 7.7度(図中B表示)の回折角(20)でのピーク高 さが、37.7度の回折角に現れるピーク高さの方が3 1. 1度の回折角に現れるピーク高さに比して相対的に 高いが、そのピーク高さの高低差は小さいX線回折パタ ーンを示すことも知られている。

[0003]

【発明が解決しようとする課題】一方、近年の切削装置の高性能化および高出力化はめざましく、かつ省力化に対する要求も強く、これに伴い、切削加工は高速化並びに高送りおよび高切込みなどの重切削化の傾向にあるが、上記の従来被覆超硬工具においては、これを構成する硬質被覆層のうち、特にA12O3層は耐酸化性と熱的安定性にすぐれ、さらに高硬度を有するが、他の構成層であるTi化合物層に比して相対的に強度が低く、かつ脆いことが原因で、例えば鋼や鋳鉄の高速切削や高速高送り切削などの苛酷な条件での切削では切刃にチッピングが発生し易く、これが原因で比較的短時間で使用寿命に至るのが現状である。

[0004]

【課題を解決するための手段】そこで、本発明者等は、上述のような観点から、被覆超硬工具の硬質被覆層を構成するA12 O3 層に着目し、これの強度および靱性向上を図るべく研究を行った結果、一般に、上記の従来被覆超硬工具の硬質被覆層を構成する粒状結晶組織を有するA12 O3 層(以下、粒状A12 O3 層と云う)は、(a) 反応ガス組成

容量%で、A1C1₃:1~20%、CO₂:0.5~30%、必要に応じてHC1:1~20%および/またはH₂S:0.05~5%、H₂:残り、

(b)反応温度:950~1050℃、

- (c) 反応圧力: 30~200torr、の条件で形成されるが、(A) Al₂O₃ 層を形成するに際して、(1) 反応開始時の条件を、
- (a) 反応ガス組成

容量%で、AICl₃:1~20%、CO₂:0.5~ 2%、H₂:残り、

- (b) 反応温度:900~950℃、
- (c) 反応圧力: 30~200torr、

とし、(2)ついで、反応圧力は変えずに、反応ガスとして H_2 Sを導入し、また CO_2 を所定時間かけて増加させ、下記温度に昇温して、

(a) 反応ガス組成

容量%で、A I C I₃ : 1~20%、CO₂ : 10~3 0%、H₂ S: 0.1~2%、H₂ : 残り、

- (b) 反応温度:950~1000℃、
- (c) 反応圧力: 30~200torr、

とした条件でA 1₂ O₃ 層を形成すると、図1の縦方向 破面の走査型電子顕微鏡による組織写真および図2の同 じく走査型電子顕微鏡による表面組織写真に見られるよ うに、「隣接相互の長さ方向の断面形状および断面寸法 に大きな相異がある縦長成長の柱状多様化多面体結晶の 並列集合組織」(以下、縦長結晶多様化組織と云う)を もったA 1₂ O₃ 層(以下、縦長多様化A 1₂ O₃ 層と 云う)が形成されるようになり、上記条件で所定層厚の 縦長多様化A 1₂ O₃ 層を形成した後で、(B)A 1₂ O₃ 層の形成条件を、

(a) 反応ガス組成

容量%で、AICl₃:1~20%、CO₂:5~15 %、H₂S:0.05~1%、H₂:残り、

- (b) 反応温度:900~950℃、
- (c) 反応圧力: 30~200torr、

とする、すなわち反応ガス中の CO_2 および H_2 Sを半減させ、かつ反応温度を相対的に下げた条件として A_1 O_3 層の形成を行なうと、同じく図1 の縦方向破面の走査型電子顕微鏡による組織写真および図3の同じく走査型電子顕微鏡による表面組織写真に見られるように、

「隣接相互の長さ方向の断面形状および断面寸法が均等化した縦長成長の柱状単一化多面体結晶の並列集合組織」(以下、縦長結晶単一化組織と云う)をもったA12〇3層(以下、縦長単一化A12〇3層と云う)が形成されるようになり、このように下側部が縦長結晶単一化組織で構成されたA12〇3層は、これら両組織の共存によって上記粒状A12〇3層に比して一段とすぐれた強度および報性をもつようになることから、これを含む硬質被覆層はすぐれた耐チッピング性を有するようになり、この場合前記A12〇3層は、通常の通り3~15μmの平均層厚で形成されるが、この平均層厚のうちの20~80%を縦長単一化A12〇3層とするのが望ましく、したがってこの結果の被覆超硬工具は、鋼や鋳鉄などの高速切削や高速

重切削などの苛酷な条件での切削にも切刃にチッピング の発生がなく、すぐれた切削性能を長期に亘って発揮す るようになるという研究結果を得たのである。

【0005】この発明は、上記の研究結果に基づいてな されたものであって、超硬基体の表面に、Al2O2層 を含む硬質被覆層、例えばTiC層、TiN層、TiC N層、TiO2層、TiNO層、およびT iCNO層からなるTi化合物層のうちの1種または2 種以上と、Al₂O₃層とで構成された硬質被覆層を8 ~25μmの平均層厚で形成してなる被覆超硬工具にお いて、前記硬質被覆層を構成するAl,O。層を、上側 部と下側部(基体側)で異なる結晶組織を有し、上記上 側部を、縦長結晶単一化組織(隣接相互の長さ方向の断 面形状および断面寸法が均等化した縦長成長の柱状単一 化多面体結晶の並列集合組織)、上記下側部を、縦長結 晶多様化組織(隣接相互の長さ方向の断面形状および断 面寸法に大きな相異がある縦長成長の柱状多様化多面体 結晶の並列集合組織)、としたAl2 Os層で構成して なる、硬質被覆層がすぐれた耐チッピング性を有する被 覆超硬工具に特徴を有するものである。

【0007】また、この発明の被覆超硬工具の硬質被覆層のうちの $A1_2O_3$ 層の上側部の層厚は、上記の通り通常の $A1_2O_3$ 層の平均層厚である $3\sim15\mu$ mの $20\sim80%$ とするのが望ましいが、これはその割合が20%未満であったり、80%を越えたりすると、相対的に $A1_2O_3$ 層の下側部の層厚が厚くなりすぎたり、あるいは薄くなりすぎたりして、これら両者の相互共存バランスがくずれるようになって硬質被覆層に所望のすぐれた耐チッピング性を確保することができない、という理由によるものであり、また、硬質被覆層の平均層厚を $8\sim25\mu$ mとしたのは、その層厚が 8μ m未満では所望のすぐれた耐摩耗性を確保することができず、一方その層厚が 25μ mを越えると、耐欠損性が低下するようになるという理由からである。

[0008]

【発明の実施の形態】つぎに、この発明の被覆超硬工具

を実施例により具体的に説明する。原料粉末として、平均粒径: 1.5μ mの細粒WC粉末、同 5μ mの粗粒WC粉末、同 5μ mの粗粒WC粉末、同 1.2μ mの(Ti, W)CN(重量比で、以下同じ、TiC/TiN/WC=24/20/56)粉末、同 1.3μ mの(Ta, Nb)C(TaC/NbC=90/10)粉末、同 1μ mのCr粉末、および同 1.2μ mのCo粉末を用意し、これら原料粉末を表1に示される配合組成に配合し、ボールミルで72時間湿式混合し、乾燥した後、 $1SO\cdot CNMG120408$ に定める形状の圧粉体にプレス成形し、この圧粉体を同じく表1に示される条件で真空焼結することにより超硬基体A~Eをそれぞれ製造した。なお、表1には、上記超硬基体A~Eの内部硬さ(ロックウエル硬さAスケール)をそれぞれ示した。

【0009】ついで、これらの超硬基体A~Eの表面に、ホーニングを施した状態で、通常の化学蒸着装置を用い、表2に示される条件にて、表3、4に示される組成および平均層厚のTi化合物層およびA12O3層からなる硬質被覆層を形成することにより本発明被覆超硬工具1~12および従来被覆超硬工具1~12をそれを制度を表現である。この結果得られた本発明被覆超硬工具1~12は、いずれもこれを構成する硬質被覆層のうちのA12O3層が、図1の本発明被覆超硬工具8の破面組織と同じ破面組織を示し、かつこれの下側部の縦長多様化A12O3層がよび上側部の縦長単一化A12O3層のそれぞれの表面に対する走査型電子顕微鏡による組織観

察および X線回折でも、図2~5 に示される本発明被覆超硬工具8の表面組織写真および X線回折パターンと同じ結果を示した。なお、図6 には、従来被覆超硬工具8の硬質被覆層を構成する粒状 A 1 2 ○3 層の X線回折パターンを示したが、これ以外の従来被覆超硬工具の A 1 2 ○3 層も同じ X線回折パターンを示した。

【0010】つぎに、上記本発明被覆超硬工具1~12 および従来被覆超硬工具1~12について、

被削材: FC300 (硬さ: HB 180) の丸棒、

切削速度: 400m/min. 、

切込み:3mm、

送り: 0.4mm/rev、

切削時間:20分、

の条件での鋳鉄の湿式連続高速高送り切削試験、並びに、

被削材: JIS・SCM440 (硬さ: HB 220) の 丸棒、

切削速度:350m/min.、

切込み:3mm.、

送り: 0.4mm/rev、

切削時間:10分、

の条件での合金鋼の乾式連続高速高送り切削試験を行い、いずれの切削試験でも切刃の逃げ面摩耗幅を測定した。これらの測定結果を表5に示した。

[0011]

【表1】

1	A H H A A)	9 1. S	92.0	91.0	90.5	89
条	保持時間 (巾 r)	-	B	-	-	ا. ت
英語	温 度 (で)	1450	1450	1400	1400	1 4 0 0
际	真空度 (torr)	0.05	0.05	0. 1	0.05	0. 1
	ა ჯ	残 (細粒)	· · · · · · · · · · · · · · · · · · ·	· · · · · · · · · · · · · · · · · · ·	獲(粗粒)	. 残 (粗粒)
成 (重量%)	, a 3	_	. 1	ľ	I	0.5
各組	(Ta, Nb) C	2	1	4	и	8
旦	(T i, W) C N	1	T.	9	ī.	. 2
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	戻	∢	æ	· U	۵	ш
	#		超	碶 基	***	

【0012】 【表2】

四	硬 質 板 觀		形成	€X	中	
ž Ze	同 序 左 7 路 舟	at B	8		反応等	民
	31 Db \ 7 A1	¥			压力 (torr)	通度(で)
0	応開始時) C $_{oldsymbol{2}}$:3%, C $_{oldsymbol{2}}$:1%, H				5 0	930
7	$(H_2 S, CO_2 ガス導入、外温核丁時)$ $A \& C \& _3 : 3\%, CO_2 : 15%, H_2 S: 1%,$	H 2:斑	and the second		5 0	1000
群長単一化Ag203	A&C&3:3%, CO2:7.5%, H2S:0	5%, 1	H ₂ :残		5.0	930
粒伏A & 2 O 3	ABC&3:5%, CO2:15%, H2S:0.	2%, нс	. д : 3%, н	五: 2	5.0	1000
TIC	TiCg4:2%, CH4:5%, H2:巍				200	1020
NIL	TiCℓ4:2%, N2:30%, H2:残				4 0 0	0 0 6
TICN	TICA4: 2%, N2: 10%, CH3 CN: 1	%, н ₂ :	## :		5.0	008
Tico	Tick4:3%, CO:2%, CH4:2%, H	2:残			200	1000
T i NO	Tice4 : 3%, CO: 1%, N2 : 15%, H	2:強			200	1000
T i CNO	Tice4:3%, CO:2%, CH4:2%, N	2:15	%, H ₂ :残		200	1000
T i O ₂	TiCs4:2%, CO2:8%, H2:現				100	1000

[0013]

【表3】

			T		T	1				_			
(mn:	第5層	ı	I	TIN (0. 2)	A & 2 03 (4 2 3 2 0)		TIN (0,.1)	T i N (0, 4)	1	T i N (0, 2)	T i N'(0, 2)	TiN (0, 3)	T N (0. 2)
(括弧内は平均層厚:um)	第 4 厘	ŀ	A & 2 0 3 (5, 0; 1, 3)	A £ 2 0 3 (4, 6; 1, 5)	0) ONO	T i N (0, 2)	A & 2 03 (4. 5 : 3. 1)	03	(0. 2	A & 2 03 (5, 8; 5, 3)	Tic (0, 3)	A # 2 03 (3. 2 : 8: 8)	A # 2 03
120			`]										
K	題 3 画	TIN (0. 2)	TICO (0, 3)	TIC (2. 5)	T I CN (6. 9)	A & 2 03 (4. 8; 2. 6)	TIN (0. 6)	T 1 02 (0. 4)	A 1 0 3 (4, 3; 4, 9)	TICO (0, 3)	A (4, 1; 7, 3)	c 3	Tic (3, 6)
蒸		3)	()	2)		6)	4)	1)	5)	2)			
誕	第2層	A & 2 03 (3, 8; 1.	TICN (6. 7	TICN (6. 2	TIN (1. 2)	TICN (4. 6	TICN (7. 4	TICN (6. 1	T I CN (8. 5	TICN (5. 2	TINO (0. 4)	TICN (6. 2)	TICN (3. 8)
		-	6	=	Ω	3)	5)	<u> </u>	33	6	9)	$\overline{}$	~
	紙	T CN (3. 1	T i N (0.	TIC (1. 1	Tic (0, 4)	TIN (1. 3	TIN (0, 5	Tic (0, 7)	TIN (0, 3	T i N (0. 6	T I GN (7.	TIC (1. 0)	TIN (1. 2)
描	記事	٧	∢	∢	В	a	ပ	ပ	٥	۵	۵	ш	ш
- E		- 1	2	က	4	ro	0	~	80	6	0	-	12
į.			₩	概	召		DS .	超	 	— <u>І</u>			

[表中、A $_2$ O $_3$ 0括四内の右側の数字は鞣摂単一化A $_2$ O $_3$ 쪔(上側部)、同左側の数字は鞣長多様化A $_2$ O $_3$ 厢(下側部)

[0014]

【表4】

			1				Γ	Γ			Γ		Γ
n ш)	第 5 届	1	I .	T i N (0, 3)	粒状A 4 2 03 (6, 8)	1	T i N (0, 3)	T i N (0, 1)	1	T i N (0, 2)	TiN (0.3)	T i N (0, 4)	TIN (0. 2)
(括弧内は平均層厚:μ m)	第4層	l	松供A 12 03 (4.8)	粒状A L ₂ O ₃ (6.4)		TIN (0, 3)	粒状A g Og (7.2)	粒状A 12 03 (8.7)	-	粒状A 62 O3 (10.6)	Tic (0, 2)	粒状Aℓ2 03 (12.7)	粒状A g O 3 (13 4)
18E						8)			5)		1)		
廊	第 3 图	T i N (0, 2)	T; CO (0, 2)	Tic (2, 2)	T : CN (7. 3)	粒状A≥2 O3 (7.	TiN (1. 0)	TiO ₂ (0.3)	粒状A≥203(9.	Tico (0, 3)	粒状Aℓ2 03 (10.1	TiN (1. 2)	TIC (3, 5)
阿	第2回	粒铁A ≠ 2 0 3 (5, 4)	Ticn (6. 3)	Ti CN (6. 0)	TiCN (0, 9)	Ti CN (4, 3)	TiCN (7. 1)	TiCN (5. 5)	TiCN (8. 7)	TiCN (5, 2)	TINO (0, 5)	TiCN (6, 3)	TICN (3. 7)
通	第一層	T i C N (3, 5)	TiN (0. 5)	Tic (1, 1)	Tic (0. 4)	TiN (1, 0)	TIN (0, 5)	Tic (0, 8)	T i N (0, 5)	T i N (0, 4)	TiCN (8, 5)	Tic (1, 1)	TiN (1, 5)
拼	記 舟	A	٨	A	80	8	ပ	Ų	۵	۵	۵	m	w
22	Ę		2	က	4	വ	9	7	8	6	1 0	-	12
質				筑	#	≉	膨	樋	康	Н	歐		

[0015]

【表5】

種	別	逃げ面摩耗幅 (mm)		種別		切 削 試 験 結 果
45	נית	鋳鉄	合金鋼	任	נימ	铸 鉄 合金鋼
	1	0. 28	0. 31		1	4. 2分で使用寿命 4. 0分で使用寿命
*	. 2	0. 27	0. 29		2	5. 3分で使用寿命 4. 9分で使用寿命
発	3	0. 25	0. 28	従	3	5.8分で使用寿命 5.3分で使用寿命
明	4	0. 25	0. 24	来	4	6. 2分で使用寿命 5. 8分で使用寿命
被	5	0. 26	0. 28	被	5	3.9分で使用寿命 3.7分で使用寿命
覆	6	0. 24	0. 22	覆	6	6. 7分で使用寿命 7. 0分で使用寿命
£2	7	0. 25	0. 26	超	7	6. 4分で使用寿命 5. 9分で使用寿命
便	8	0. 24	0. 25	硬.	8	6.8分で使用寿命 6.5分で使用寿命
I	9	0. 23	0. 22	エ	9	7. 3分で使用寿命 6. 9分で使用寿命
具	1 0	0. 21	0. 23	具	10	8. 4分で使用寿命 7. 5分で使用寿命
~	11	0. 20	0. 22		11	8. 6分で使用寿命 7. 9分で使用寿命
	1 2	0. 18	0. 16		12	9.0分で使用寿命 9.4分で使用寿命

(使用寿命はいずれも切刃に発生したチッピングが原因)

[0016]

【発明の効果】表 $3\sim5$ に示される結果から、硬質被覆層におけるA 1_2 O_3 層の上側部が縦長単一化A 1_2 O_3 層で、下側部が縦長多様化A 1_2 O_3 層で構成された本発明被覆超硬工具 $1\sim1$ 2 は、これの全体が粒状 A 1_2 O_3 層で構成された従来被覆超硬工具 $1\sim1$ 2 に比して、前記縦長単一化A 1_2 O_3 層と縦長多様化A 1_2 O_3 層の共存作用で、前記粒状 A 1_2 O_3 層に比してきわめて高い強度および靱性をもつようになることから、苛酷な切削条件となる鋳鉄および鋼の高速高送り切削でも切刃にチッピングの発生なく、すぐれた切削性能を長期に亘って発揮するのに対して、従来被覆超硬工具 $1\sim1$ 2 においては、粒状 A 1_2 O_3 層が原因で切刃にチッピングの発生は避けられず、いずれもこれが原因で比較的

短時間で使用寿命に至ることが明らかである。上述のように、この発明の被覆超硬工具は、これを構成する硬質被覆層のうちのA12O3層を縦長単一化A12O3層の上側部と縦長多様化A12O3層の下側部で構成することによってすぐれた耐チッピング性を有するようにしたものであり、したがって鋼や鋳鉄などの通常の条件での連続切削や断続切削は勿論のこと、高速切削や高速高送りおよび高速高切込み切削などの苛酷な条件での切削に用いた場合にも切刃にチッピングの発生なく、長期に亘ってすぐれた切削性能を発揮し、切削装置の高性能化および高出力化に十分に対応でき、かつ省力化にも寄与するものである。

【図面の簡単な説明】

【図1】本発明被覆超硬工具8の縦方向破面の走査型電

子顕微鏡による組織写真(倍率:4000倍)である。 【図2】本発明被覆超硬工具8の硬質被覆層を構成する A1₂O₃層における縦長多様化A1₂O₃層の走査型 電子顕微鏡による表面組織写真(倍率:5000倍)である。

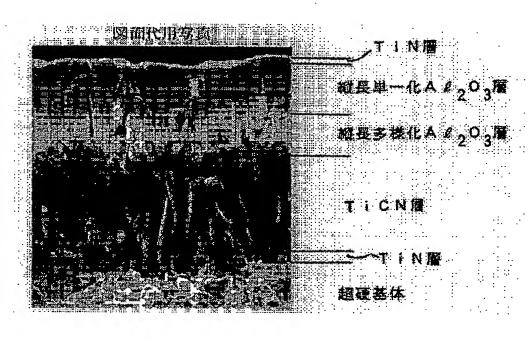
【図3】本発明被覆超硬工具8の硬質被覆層を構成する A 1₂ O₃ 層における縦長単一化A 1₂ O₃ 層の走査型 電子顕微鏡による表面組織写真(倍率:5000倍)である。

【図4】本発明被覆超硬工具8の硬質被覆層を構成する A_1_2 O_3 層における縦長多様 A_1_2 O_3 層のX線回 折パターンを示す図である。

【図5】本発明被覆超硬工具8の硬質被覆層を構成する $A1_2O_3$ 層における縦長単一 $(A1_2O_3)$ 層のX線回 折パターンを示す図である。

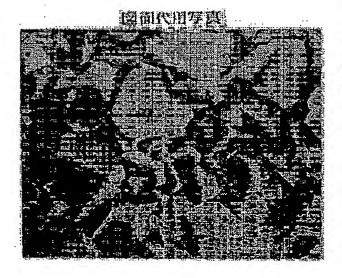
【図6】従来被覆超硬工具8の硬質被覆層を構成する粒 状A1₂0₈層のX線回折パターンを示す図である。

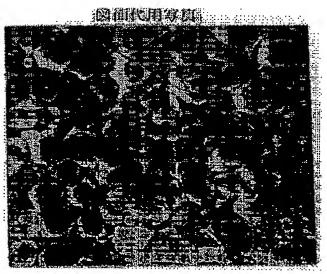
【図1】



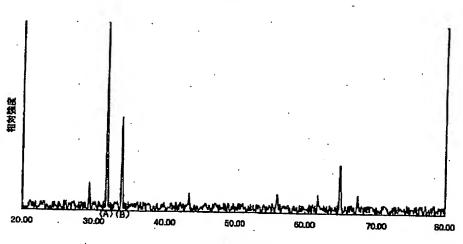
【図2】

【図3】



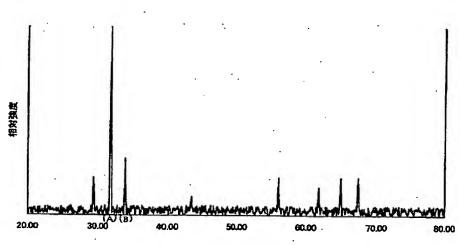






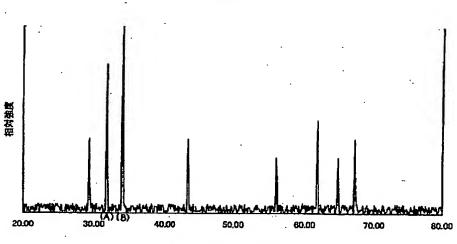
回折角(28)

【図5】



回折角(28)





回折角(20)

フロントページの続き

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OSADA AKIRA

OOSHIKA TAKATOSHI

(54) CUTTING TOOL MADE OF SURFACE COATED CEMENTED CARBIDE ALLOY WITH HARD COATING LAYER HAVING EXCELLENT ANTICHIPPING PROPERTY

(57)Abstract:

PROBLEM TO BE SOLVED: To provide a cutting edge without generating any chipping (micro defect) while performing an excellent cutting performance in a long period even in the case for applying the high speed cutting and the high heavy cutting such as high feed and deep cut steel or cast iron.

SOLUTION: This tool is composed by evaporating chemically or physically a hard quality covered layer by one sort or two or more sorts of hard quality covered layers including an Al2O3 layer, such as a Ti compound layer consisting of TiO layer, TiN layer, TiCN layer, TiO2 layer, TiCO layer, TiNO layer, and TiCNO layer, and an Al2O3 layer, at an average layer thickness of 8 to 25 μ m. In this case, the Al2O3 layer to compose the hard quality covered layer is composed of the Al2O3 layers having different crystal compositions, at the upper side and the lower side (base body side). That is, the above upper side is composed of an Al2O3 layer having a parallel aggregate formation of a longitudinal growth of columnar single polyhedron crystal uniforming the section form and section size in the longitudinal direction neighboring each other, while the above lower side is composed of an Al2O3 layer having a parallel aggregate formation of a longitudiral growth of columnar various polyhedron crystal having a large difference in the section form and the section size in the longitudinal direction neighboring each other.

LEGAL STATUS

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the examiner's decision of rejection or application converted registration]

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[Patent number]

3331929

[Date of registration]

26.07.2002

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CLAIMS

[Claim(s)]

[Claim 1] The hard enveloping layer which contains an aluminum—oxide layer on the front face of a tungsten carbide radical cemented carbide base is set by 8–25-micrometer average thickness to chemical vacuum deposition and/or the cutting tool made from surface coating cemented carbide which comes to carry out a physical vapor deposition. It has the crystalline structure which is different in the top section and the bottom section (base side) in the aluminum—oxide layer which constitutes said hard enveloping layer. The juxtaposition texture of the pillar—shaped simplification polyhedron crystal of the longwise growth for which the cross—section configuration and cross—section dimension of the die—length direction between contiguity equated the above—mentioned top section. The cutting tool made from surface coating cemented carbide which has the chipping—proof nature excellent in the hard enveloping layer characterized by constituting the above—mentioned bottom section from an aluminum oxide layer made into the juxtaposition texture of the pillar—shaped diversification polyhedron crystal of the longwise growth which has big difference in the cross—section configuration and cross—section dimension of the die—length direction between contiguity.

[Claim 2] The cutting tool made from surface coating cemented carbide characterized by providing the following One sort in Ti compound layer which becomes the front face of a tungsten carbide radical cemented carbide base from the carbide layer, the nitride layer, the carbon nitride layer, the oxide layer, the carbonic acid ghost layer, nitric oxide layer, and charcoal nitric oxide layer of Ti, or two sorts or more The hard enveloping layer which consisted of aluminum-oxide layers is set by 8-25-micrometer average thickness to chemical vacuum deposition and/or the cutting tool made from surface coating cemented carbide which comes to carry out a physical vapor deposition. It has the crystalline structure which is different in the top section and the bottom section (base side) in the aluminum-oxide layer which constitutes said hard enveloping layer. The juxtaposition texture of the pillar-shaped simplification polyhedron crystal of the longwise growth for which the cross-section configuration and cross-section dimension of the die-length direction between contiguity equated the above-mentioned top section, Chipping-proof nature excellent in the hard enveloping layer characterized by constituting the above-mentioned bottom section from an aluminum oxide layer made into the juxtaposition texture of the pillar-shaped diversification polyhedron crystal of the longwise growth which has big difference in the cross-section configuration and cross-section dimension of the die-length direction between contiguity

[Claim 3] By the X diffraction using Cukappa alpha rays over each front face of the top section of the above-mentioned aluminum-oxide layer, and the bottom section as a line source The peak height in the angle of diffraction (2theta) which are 31.1 degrees in which the peak of an aluminum oxide appears, and 37.7 degrees The cutting tool made from surface coating cemented carbide given in above-mentioned claims 1 or 2 characterized by the peak height which appears in the angle of diffraction of 31.1 degrees as compared with the peak height which appears in the angle of diffraction of 37.7 degrees being relatively higher, and the difference of elevation of the peak height being large.

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DETAILED DESCRIPTION

[Detailed Description of the Invention]

[Field of the Invention] Also when this invention had the chipping-proof nature excellent in the hard enveloping layer, therefore it is steel, the high speed cutting of cast iron, and a high speed and it is used on severe conditions, such as deep cuts of high delivery, high infeed, etc., there is no generating of a chipping (minute chip) in a cutting edge, and it relates to the cutting tool made from surface coating cemented carbide (henceforth a coated carbide tool) which demonstrates the cutting-ability ability which continued and was excellent in the long period of time. [0002]

[Description of the Prior Art] Generally conventionally on the front face of a tungsten carbide radical cemented carbide base (henceforth a superhard base) The hard enveloping layer containing an aluminum-oxide (aluminum 203 shows hereafter) layer, For example, the carbide (TiC shows hereafter) layer of Ti, a nitride (similarly TiN shows hereafter) layer, A carbon nitride (TiCN shows hereafter) layer, an oxide (TiO2 shows hereafter) layer, One sort in a carbonation object (TiCO shows hereafter) layer, a nitric oxide (TiNO shows hereafter) layer, and Ti compound layer that consists of a charcoal nitric oxide (TiCNO shows hereafter) layer, or two sorts or more, aluminum 203 Chemical vacuum deposition and/or the coated carbide tool which comes to carry out a physical vapor deposition are known for 10-25-micrometer average thickness in the hard enveloping layer which consisted of layers, and it is also known that this coated carbide tool is used for continuation cutting and intermittence cutting of steel, cast iron, etc. Moreover, aluminum 203 which constitutes the above-mentioned hard enveloping layer It is also known that a layer has the granular crystalline structure and it is the above aluminum 203 further. By the X diffraction using Cukappa alpha rays over the front face of a layer as a line source It is aluminum 203 so that it may be illustrated by the X diffraction pattern of drawing 6. The peak height in the angle of diffraction (2theta) of 31.1 degrees (A display in drawing) and 37.7 degrees (B display in drawing) in which a peak appears As compared with the peak height which appears in the angle of diffraction whose direction of the peak height which appears in the angle of diffraction of 37.7 degrees is 31.1 degrees, although it is relatively high, it is also known that the difference of elevation of the peak height shows a small X diffraction pattern. [0003]

[Problem(s) to be Solved by the Invention] On the other hand, although high-performance-izing and the high increase in power of cutting equipment in recent years are remarkable, and the demand to laborsaving is also strong and cutting is in the inclination of deep-cuts-izing of high delivery, high infeed, etc. in connection with this at a high speed list It is especially aluminum 203 among the hard enveloping layers which constitute this in the above-mentioned conventional coated carbide tool. Although a layer is excellent in oxidation resistance and thermal stability and it has a high degree of hardness further The present condition is that are because reinforcement is relatively weak low, for example, are easy to generate a chipping in a cutting edge as compared with Ti compound layer which are other configuration layers in cutting on severe conditions, such as steel, high speed cutting of cast iron, and high-speed quantity delivery cutting, and this results in a use life comparatively for a short time owing to.

[0004]

[Means for Solving the Problem] Then, this invention person etc. is aluminum 203 which constitutes the hard enveloping layer of a coated carbide tool from the above viewpoints. Its attention is paid to a layer, aluminum 203 which generally has the granular crystalline structure which constitutes the hard enveloping layer of the above-mentioned conventional coated carbide tool as a result of inquiring in order to aim at reinforcement of this, and improvement in toughness Layer It is (a) reactant gas presentation capacity %. (— hereafter granular — it is called aluminum2 O3 layer —) — AICI3: 1 – 20%, and CO2: 0.5 – 30%, and the need — responding — HCI: — 1 – 20% and/or H2 S:0.05–5%, and H2: Although it remains and is formed on condition that (b) reaction temperature:950–1050 degree C and (c) reaction pressure:30 – 200torr**

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TECHNICAL FIELD

[Field of the Invention] Also when this invention had the chipping-proof nature excellent in the hard enveloping layer, therefore it is steel, the high speed cutting of cast iron, and a high speed and it is used on severe conditions, such as deep cuts of high delivery, high infeed, etc., there is no generating of a chipping (minute chip) in a cutting edge, and it relates to the cutting tool made from surface coating cemented carbide (henceforth a coated carbide tool) which demonstrates the cutting-ability ability which continued and was excellent in the long period of time.

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PRIOR ART

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EFFECT OF THE INVENTION

[Effect of the Invention] aluminum 203 in the hard enveloping layer from the result shown in Tables 3-5 The top section of a layer is the longwise simplification aluminum 203. It is a layer, The bottom section is the longwise diversification aluminum 203, this invention coated carbide tools 1-12 which consisted of layers This whole is granular aluminum 203. It compares with coated carbide tools 1-12 conventionally which consisted of layers, and is said longwise simplification aluminum 203. A layer and longwise diversification aluminum 203 In a coexistence operation of a layer Said granular aluminum 203 From coming to have very high reinforcement and toughness as compared with a layer As opposed to continuing and demonstrating the cutting-ability ability which high-speed quantity delivery cutting of the cast iron used as severe cutting conditions and steel does not have generating of a chipping in a cutting edge, either, and excelled [cutting edge] in it at a long period of time It sets to coated carbide tools 1-12 conventionally, and is granular aluminum 203. Owing to a layer, generating of a chipping is not avoided by the cutting edge but it is clear to it all to result in a use life comparatively for a short time owing to this. As mentioned above, the coated carbide tool of this invention is aluminum 203 of the hard enveloping layers which constitute this. It is the longwise simplification aluminum 203 about a layer. The top section of a layer, and longwise diversification aluminum 203 It constitutes from the bottom section of a layer. It is made [therefore] to have the outstanding chipping-proof nature. Not to mention continuation cutting on the usual conditions and intermittence cutting of steel, cast iron, etc. Also when it uses for cutting on severe conditions, such as high speed cutting, and high-speed quantity delivery, high-speed quantity infeed cutting, there is no generating of a chipping in a cutting edge, the cutting-ability ability which continued and was excellent in the long period of time is demonstrated, and it can fully respond to highperformance-izing and the high increase in power of cutting equipment, and contributes also to laborsaving.

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TECHNICAL PROBLEM

[Problem(s) to be Solved by the Invention] On the other hand, although high-performance-izing and the high increase in power of cutting equipment in recent years are remarkable, and the demand to laborsaving is also strong and cutting is in the inclination of deep-cuts-izing of high delivery, high infeed, etc. in connection with this at a high speed list It is especially aluminum 203 among the hard enveloping layers which constitute this in the above-mentioned conventional coated carbide tool. Although a layer is excellent in oxidation resistance and thermal stability and it has a high degree of hardness further The present condition is that are because reinforcement is relatively weak low, for example, are easy to generate a chipping in a cutting edge as compared with Ti compound layer which are other configuration layers in cutting on severe conditions, such as steel, high speed cutting of cast iron, and high-speed quantity delivery cutting, and this results in a use life comparatively for a short time owing to.

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MEANS

[Means for Solving the Problem] Then, this invention person etc. is aluminum 203 which constitutes the hard enveloping layer of a coated carbide tool from the above viewpoints. Its attention is paid to a layer. aluminum 203 which generally has the granular crystalline structure which constitutes the hard enveloping layer of the above-mentioned conventional coated carbide tool as a result of inquiring in order to aim at reinforcement of this, and improvement in toughness Layer It is (a) reactant gas presentation capacity %. (-- hereafter granular -- it is called aluminum2 O3 layer --) -- AICl3 : 1 - 20%, and CO2 : 0.5 - 30%, and the need -responding -- HCI: -- 1 - 20% and/or H2 S:0.05-5%, and H2: Although it remains and is formed on condition that (b) reaction temperature:950-1050 degree C and (c) reaction pressure:30 -200torr** (A) aluminum 203 It faces forming a layer. The conditions at the time of (1) reaction initiation by (a) reactant gas presentation capacity % AICI3: 1 - 20%, and CO2: 0.5 - 2%, and H2: remaining -- (b) reaction temperature:900-950 degree C and (c) reaction pressure:30 - 200torr -- carrying out -- (2) -- subsequently Without changing, reaction pressure introduces H2 S as reactant gas, and is CO2. Predetermined-time-apply, it is made to increase, a temperature up is carried out to the following temperature, and it is (a) reactant gas presentation capacity %. AICI3 :.1 - 20%, CO2 : 10 - 30%, H2 S:0.1-2%, H2 : The remainder, (c) reaction pressure: It is aluminum 203 at the conditions set to 30 - 200torr. If a layer is formed, [(b) reaction temperature: 950 -] [1000 degrees C and] So that the organization photograph by the scanning electron microscope of the lengthwise direction fracture surface of drawing 1 and the surface organization photograph of <u>drawing 2</u> similarly according to a scanning electron microscope may see "The juxtaposition texture of the pillar-shaped diversification polyhedron crystal of the longwise growth which has big difference in the cross-section configuration and cross-section dimension of the die-length direction between contiguity"

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DESCRIPTION OF DRAWINGS

[Brief Description of the Drawings]

[Drawing 1] It is an organization photograph (scale factor: 4000 times) by the scanning electron microscope of the lengthwise direction fracture surface of this invention coated carbide tool 8. [Drawing 2] aluminum 203 which constitutes the hard enveloping layer of this invention coated carbide tool 8 Longwise diversification aluminum 203 in a layer It is a surface organization photograph (scale factor: 5000 times) by the scanning electron microscope of a layer. [Drawing 3] aluminum 203 which constitutes the hard enveloping layer of this invention coated carbide tool 8 Longwise simplification aluminum 203 in a layer It is a surface organization photograph (scale factor: 5000 times) by the scanning electron microscope of a layer. [Drawing 4] aluminum 203 which constitutes the hard enveloping layer of this invention coated carbide tool 8 Longwise diversification aluminum 203 in a layer It is drawing showing the X diffraction pattern of a layer.

[Drawing 5] aluminum 203 which constitutes the hard enveloping layer of this invention coated carbide tool 8 Longwise simplification aluminum 203 in a layer It is drawing showing the X diffraction pattern of a layer.

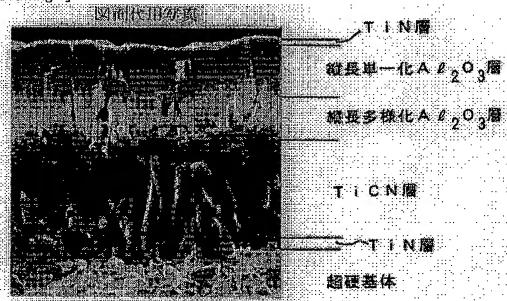
[Drawing 6] Granular aluminum 203 which constitutes the hard enveloping layer of a coated carbide tool 8 conventionally It is drawing showing the X diffraction pattern of a layer.

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DRAWINGS

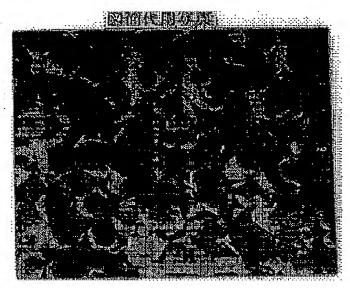
[Drawing 1]

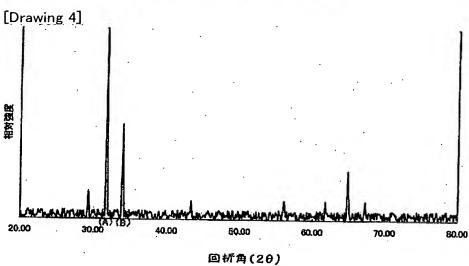


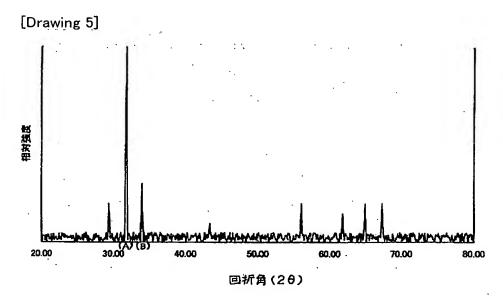
[Drawing 2]



[Drawing 3]







[Drawing 6]

